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# (54) IMPACT MODIFIER, PROCESS FOR PRODUCTION, AND RESIN COMPOSITION CONTAINING THE SAME

(57) The present invention relates to an impact strength modifier capable of imparting both excellent weathering resistance and impact resistance, and to a resin composition comprising the impact strength modifier and a thermoplastic resin, particularly a rigid vinyl chloride resin. Since this impact strength modifier comprises, as a principal component, an acrylic rubber graft copolymer which is obtained by graft polymerization of a polyalkyl (meth)acrylate (A) rubber containing a plu-

rality of acrylic rubber components composed of a specific monomer as a principal component, and a vinyl monomer (B), it is made possible to improve the impact resistance, particularly impact resistance at low temperatures, of the resin and to satisfactorily maintain the weathering resistance by adding a small amount of the impact strength modifier to the resin. This impact strength modifier is particularly suited for incorporation into a rigid vinyl chloride resin.

### Description

### **TECHNICAL FIELD**

[0001] The present invention relates to an impact strength modifier capable of imparting both excellent weathering resistance and impact resistance, and to a resin composition comprising the impact strength modifier and a thermoplastic resin, particularly a rigid vinyl chloride resin.

[0002] This application is based on a Japanese Patent Application No. Hei 10-198882 filed in Japan, the contents of which are incorporated herein by reference.

### BACKGROUND ART

[0003] Thermoplastic resins, particularly vinyl chloride resins, are widely used resins, but have such drawbacks as poor impact resistance. To improve the impact resistance, various methods have been suggested.

[0004] For example, there has been suggested a method of using a mixture prepared by mixing an MBS resin, which is obtained by graft polymerization of a butadiene rubber polymer with methyl methacrylate, styrene or acrylonitrile, with a vinyl chloride resin. However, when using the MBS resin in combination with the vinyl chloride resin, the impact resistance is improved but the weathering resistance is lowered. Therefore, a molded article of the vinyl chloride resin incorporating the MBS resin has such the drawback that the impact resistance is drastically lowered when using it outdoors. It is considered that the weathering resistance is lowered by deterioration of a butadiene unit constituting the MBS resin due to ultraviolet light.

[0005] Thus, there has been suggested a method of providing a vinyl chloride resin with both weathering resistance and impact resistances by using, as an impact strength modifier, resins obtained by graft polymerization of a crosslinked alkyl (meth) acrylate rubber polymer comprising an alkyl (meth) acrylate and a crosslinking agent, and methyl methacrylate, styrene or acrylonitrile in Japanese Unexamined Patent Publication, First Publication No. Sho 51-28117. When using such an acrylic graft copolymer as the impact strength modifier, the resulting molded article is superior in weathering resistance and is less susceptible to lowering of the impact resistance. However, there is the problem that a large amount of the acrylic graft copolymer must be added to impact the impact resistance and properties which can provide impact strength at low temperatures, are poor as compared with the case of using the MBS resin as the impact strength modifier.

[0006] As the another method of improving the impact resistance of the thermoplastic resin, for example, a method of using a multi-layer composite interpolymer having a two-mode distribution, comprising an elastomer inner phase and a rigid non-elastomer outer phase is suggested in Japanese Unexamined Patent Publication, First Publication No. Sho 63-270715 and Japanese Unexamined Patent Publication, First Publication No. Sho 63-270716.

[0007] The impact strength modifier further includes, for example, a core/shell type impact strength modifier comprising a core section having rubber elasticity and a shell section made of a rigid thermoplastic resin component, which is suggested in U.S. Patent No. 3,678,133, and an impact strength modifier obtained by polymerizing a specific alkyl acrylate after subjecting it to a homogenization treatment to prepare a rubber polymer, and polymerizing a vinyl monomer in the presence of the rubber polymer without adding an initiator, which is suggested in Japanese Unexamined Patent Publication, First Publication No. Sho 63-33764.

[0008] However, the impact resistance of the resin could not be improved sufficiently even when using any of these impact strength modifiers.

### DISCLOSURE OF THE INVENTION

[0009] An object of the present invention is to provide an impact strength modifier, which improves the impact resistance, particularly impact resistance at low temperatures, of the resin and satisfactorily maintains the weathering resistance of the resulting molded article by adding a small amount of the impact strength modifier.

[0010] The impact strength modifier of the present invention can improve the impact resistance, particularly the impact resistance at low temperatures, of the resin and satisfactorily maintain the weathering resistance of the resin by adding a small amount of the impact strength modifier to the resin. Accordingly, a molded article of a resin composition incorporating this impact strength modifier is superior in impact resistance and also satisfactorily maintains its weathering resistance such as fade resistance or impact resistance. This impact strength modifier is particularly suited for incorporation into a vinyl chloride resin.

### BRIEF DESCRIPTION OF THE INVENTION

[0011] Fig. 1 is a graph showing a particle size distribution (number distribution) of a polyalkyl (meth) acrylate rubber

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of the Examples of the present invention.

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[0012] Fig. 2 is a graph showing a particle size distribution (weight distribution) of a polyalkyl (meth) acrylate rubber of the Examples of the present invention.

## 5 BEST MODE FOR CARRYING OUT THE INVENTION

[0013] The present invention will be described in detail below.

[0014] The polyalkyl (meth) acrylate rubber (A) used in the present invention contains, as a principal component, an acrylic rubber (A1) component and an acrylic rubber (A2) component. The acrylic rubber (A1) component contains, as a constituent component, at least one of a (meth) acrylate of an alcohol having a branched side chain or an alcohol having an alkyl group with 13 or more carbon atoms and a (meth)acrylate having a hydroxyl group, a methoxy group or an ethoxy group. The acrylic rubber (A2) component contains n-butyl acrylate as a constituent component.

[0015] The (meth)acrylate of the alcohol having a branched side chain or the alcohol having an alkyl group with 13 or more carbon atoms includes, for example, 2-ethylhexyl acrylate, tridecyl methacrylate, stearyl methacrylate, tridecyl acrylate, stearyl acrylate, or cetyl methacrylate. The (meth)acrylate having a hydroxyl group, a methoxy group or an ethoxy group includes, for example, ethoxyethyl acrylate, methoxypropylene glycol acrylate, or 4-hydroxybutyl acrylate. Among these (meth)acrylates, 2-ethylhexyl acrylate, ethoxyethyl acrylate, methoxypropylene glycol acrylate, 4-hydroxybutyl acrylate, tridecyl methacrylate, or stearyl methacrylate is preferably used, and 2-ethylhexyl acrylate is particularly preferably used.

[0016] The polyalkyl (meth)acrylate rubber (A) is preferably composed of the acrylic rubber (A1) component containing at least one of the above monomers as a constituent component and the acrylic rubber (A2) component containing n-butyl acrylate as a constituent component because the resulting impact strength modifier exhibits high impact resistance.

[0017] The polyalkyl (meth) acrylate rubber (A) is more preferably composed of the acrylic rubber (A1) component containing at least one of 2-ethylhexyl acrylate, tridecyl methacrylate and stearyl methacrylate as a constituent component and the acrylic rubber (A2) component containing n-butyl acrylate as a constituent component because the resulting impact strength modifier exhibits a high impact resistance at any temperature. The acrylic rubber (A1) component is most preferably composed of 2-ethylhexyl acrylate because the resulting impact strength modifier exhibits excellent impact resistance from a low temperature range.

[0018] With respect to the proportion of the acrylic rubber (A1) component and acrylic rubber (A2) component in the polyalkyl (meth)acrylate rubber (A), the amount of the acrylic rubber (A1) component is preferably within a range from 5 to 95% by weight and that of the acrylic rubber (A2) component is preferably within a range from 95 to 5% by weight because the resulting impact strength modifier exhibits higher impact resistance.

[0019] The polyalkyl (meth) acrylate rubber (A) preferably has two or more glass transition temperatures at 10°C or lower and, furthermore, the glass transition temperature (Tg1) derived from the acrylic rubber (A1) component is preferably lower than the glass transition temperature (Tg2) derived from the acrylic rubber (A2) component. It is preferable for the glass transition temperature of the polyalkyl (meth)acrylate rubber (A) to meet these conditions because the resulting impact strength modifier with exhibit higher impact resistance.

[0020] The glass transition temperature of the polymer is measured as a transition point of Tan  $\delta$  which is measured by a dynamic mechanical properties analyzer (hereinafter referred to as DMA), Generally, the polymer obtained from the monomer has an intrinsic glass transition temperature and one transition point was observed in the case of a homopolymer (random copolymer composed of a single of plural components), while plural intrinsic transition points are observed in case of a mixture composed of plural components, or a composite polymer.

[0021] If the polymer is composed of two components, two transition points are observed by the measurement. Although two peaks are observed in the Tan  $\delta$  curve measured by DMA, the respective peaks are near to with each other and are sometimes observed as a peak with a shoulder if the two components drastically differ in composition ratio and have almost the same transition temperature. However, it can be differentiated from a simple one-peak curve. [0022] The polyalkyl (meth)acrylate rubber (A) preferably contains a monomer having two or more unsaturated bonds in its molecule in the amount of 20% by weight or less. Such a monomer serves as a crosslinking agent or a graft cross agent.

[0023] Examples of the crosslinking agent include ethylene glycol dimethacrylate, propylene glycol dimethacrylate, 1,3-butylene glycol dimethacrylate, 1,4-butylene glycol dimethacrylate, divinylbenzene, and silicone such as polyfunctional methacrylic group-modified silicone.

[0024] Examples of the graft cross agent include allyl methacrylate, triallyl cyanurate, and triallyl isocyanurate. Allyl methacrylate can also be used as the crosslinking agent. These crosslinking agents and graft cross agents are used alone or in combination thereof.

[0025] These crosslinking agents and graft cross agents are preferably contained in the acrylic rubber (A1) component and acrylic rubber (A2) component, which constitute the polyalkyl (meth) acrylate rubber (A), in an amount within

a range from 0 to 20% by weight. Furthermore, they are preferably contained in the acrylic rubber (A1) component and acrylic rubber (A2) component in the amount within a range from 0.1 to 20% by weight.

[0026] When these crosslinking agents and graft cross agents are contained in the amount within the above range, the resulting impact strength modifier exhibits more excellent impact resistance and the resin incorporated with this impact strength modifier is superior in processability. On the other hand, when the crosslinking agents and graft cross agents are contained in the polyalkyl (meth) acrylate rubber (A) in an amount larger than 20% by weight, the resulting impact strength modifier sometimes does not exhibit excellent impact resistance.

[0027] In the monomer constituting the polyalkyl (meth)acrylate rubber (A), various vinyl monomers, for example, an aromatic alkenyl compound such as styrene,  $\alpha$ -methylstyrene or vinyl toluene; cyanated vinyl compound such as acrly-onitrile or methacrylonitrile; methacrylic acid-modified silicone; and fluorine-containing vinyl compound may be added in an amount of 30% by weight or less.

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[0028] Regarding the polyalkyl (meth)acrylate rubber (A), the number distribution of the particle size is a double dispersion distribution having at least two peaks within a range from 0.05 to 0.4  $\mu$ m and, furthermore, not less than 70% by weight of the polyalkyl (meth)acrylate rubber (A) is occupied by particles having a particle size within a range from 0.05 to 0.4  $\mu$ m. The impact strength modifier obtained by using the polyalkyl (meth)acrylate rubber (A), which exhibits such a particle size distribution, is preferred because it provides excellent impact resistance only by adding a small amount of it. Furthermore, when the number distribution of the particle size of the polyalkyl (meth)acrylate rubber (A) has at least one peak within a range from 0.05 to 0.15  $\mu$ m and a range from 0.15 to 0.4  $\mu$ m, an impact strength modifier having a more excellent impact resistance can be obtained.

[0029] As used herein, the number distribution refers to a distribution wherein the proportion of the number of particles, the particle size of which is within a microspace between a certain particle size  $d_p$  and  $d_p + \Delta d_p$ , to all particles is represented by a percentage.

[0030] The method of preparing the polyalkyl (meth) acrylate rubber (A) is preferably a method of polymerizing a monomer containing, as a constituent component, at least one of a (meth)acrylate of an alcohol having a branched side chain or an alcohol having an alkyl group with 13 or more carbon atoms and a (meth)acrylate having a hydroxyl group, a methoxy group or an ethoxy group to obtain a latex of an acrylic rubber (A1) component, adding a monomer containing n-butyl acrylate constituting an acrylic rubber (A2) component in the latex of the acrylic rubber (A1) component, and polymerizing the latex impregnated with the monomer in the presence of a radical polymerization initiator. With the progress of the polymerization, a latex of the polyalkyl (meth)acrylate rubber (A) combined with the acrylic rubber (A1) component and the acrylic rubber (A2) component is obtained.

[0031] Alternatively, the polyalkyl (meth)acrylate rubber(A) may be prepared by subjecting n-butyl acrylate to dropwise addition polymerization in the presence of the acrylic rubber (A1) component. The resulting polyalkyl (meth)acrylate rubber (A) may be enlarged by an acid or base.

[0032] These polymerization methods are not specifically limited, but an emulsion polymerization method or, if necessary, a forced emulsion polymerization method is usually used.

[0033] As the acrylic rubber (A1) component, at least one of 2-ethylhexyl acrylate, tridecyl methacrylate and stearyl methacrylate is more preferably contained as a constituent component. Since these monomers have poor water solubility, the monomer is preferably polymerized by using the forced emulsion polymerization method if the acrylic rubber (A1) component is prepared by using the monomer. Even when a monomer, which generates a large amount of scales by a conventional emulsion polymerization method, is polymerized, the polymerization proceeds smoothly by using the forced emulsion polymerization method and a polyalkyl (meth)acrylate rubber (A) having the desired particle size is easily obtained.

[0034] As the emulsifier and dispersion stabilizer, known surfactants such as anionic, nonionic and cationic surfactants can be used. If necessary, a mixture thereof can be used, but an emulsifier having a large micelle forming capability and an emulsifier having a small micelle forming capability are preferably used in combination.

[0035] The method of adding the monomer containing n-butyl acrylate constituting the acrylic rubber (A2) component to the acrylic rubber (A1) component obtained by the forced emulsion polymerization method, and polymerizing the acrylic rubber (A1) component impregnated with the monomer is a preferable method which is effective to obtain an acrylic graft copolymer as a principal component of the impact strength modifier.

[0036] If the polyalkyl (meth)acrylate rubber (A) is obtained by polymerizing n-butyl acrylate constituting the acrylic rubber (A2) component in the presence of the acrylic rubber (A1) component obtained by the forced emulsion polymerization method and the number distribution of the particle size of the polyalkyl (meth)acrylate rubber (A) is a double dispersion distribution having at least two peaks within a range from 0.05 to 0.4 µm and not less than 70% by weight of the polyalkyl (meth) acrylate rubber (A) consists of particles having a particle size within a range from 0.05 to 0.4 µm, the impact strength modifier obtained by using this polyalkyl (meth) acrylate rubber (A) provides excellent impact resistance as compared with an impact strength modifier which is obtained from the polyalkyl (meth)acrylate rubber (A) obtained by mixing plural particles having different particle sizes, which is particularly preferred.

[0037] Examples of the vinyl monomer (B) to be graft-polymerized with the polyalkyl (meth)acrylate rubber (A) include

various vinyl monomers, for example, an aromatic alkenyl compound such as styrene,  $\alpha$ -methylstyrene, or vinyl toluene; methacrylate such as methyl methacrylate, or 2-ethylhexyl methacrylate; acrylate such as methyl acrylate, ethyl acrylate or n-butyl acrylate, and cyanated vinyl compound such as acrionitrile or methacrylonitrile. These vinyl monomers are used alone or in combination. If necessary, chain transfer agents and crosslinking agents may be used.

[0038] With respect to the proportion of the polyalkyl (meth) acrylate rubber (A) to the vinyl monomer (B) in the acrylic rubber graft copolymer, the amount of the polyalkyl (meth)acrylate rubber (A) is preferably within a range from 5 to 95% by weight, more preferably from 50 to 95% by weight, still more preferably from 80 to 95% by weight, and particularly preferably from 82 to 95% by weight, while the amount of the vinyl monomer (B) is preferably within a range from 95 to 5% by weight, more preferably from 50 to 5% by weight, still more preferably from 20 to 5% by weight, and particularly preferably from 18 to 5% by weight.

[0039] When the amount of the vinyl monomer (B) is smaller than 5% by weight, the dispersibility becomes poor in the resin composition and the processability is sometimes lowered when the resulting impact strength modifier is dispersed in the resin. On the other hand, when the amount exceeds 95% by weight, the resulting impact strength modifier sometimes does not exhibit sufficient impact resistance.

[0040] The acrylic graft copolymer can be obtained by single- or multi-stage polymerization using a method of adding the vinyl monomer (B) to the latex of the polyalkyl (meth)acrylate and subjecting the mixture to the radical polymerization.

[0041] The impact strength modifier is obtained by charging an acrylic rubber graft copolymer latex in hot water, in which an acid such as sulfuric acid or hydrochloric acid, or a metal salt such as calcium chloride, calcium acetate or magnesium sulfate is dissolved, followed by salting-out, solidification, separation and further recovering. At this time, salts such as sodium carbonate and sodium sulfate may be used in combination. It may be obtained by a direct drying method such as spray-drying method.

[0042] The impact strength modifier thus obtained can be incorporated into various resins, including a rigid vinyl chloride resin (PVC resin).

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[0043] Examples of the resin other than the rigid vinyl chloride resin (PVC resin) include olefin resin (olefin resin) such as semi-rigid and soft vinyl chloride resin (PVC resin), polypropylene (PP), or polyethylene (PE); styrene resin (St resin) such as polystyrene (PS), high-impact polystyrene (HIPS), (meth)acrylate-styrene copolymer (MS), styrene-acrylonitrile copolymer (SAN), styrene-maleic anhydride copolymer (SMA), ABS, ASA, or AES; acrylic resin (Ac resin) such as methyl polymethacrylate (PMMA); polycarbonate resin (PC resin); polyamide resin (PA resin); polyester resin (PEs resin) such as polyethylene terephthalate (PET), or polybutylene terephthalate (PBT); engineering plastics such as (modified) polyphenylene ether resin (PPE resin), polyoxymethylene resin (POM resin), polysulfon resin (PSO resin), polyarylate resin (PAr resin), polyphenylene resin (PPS resin), or thermoplastic polyurethane resin (PU resin); thermoplastic elastomer (TPE) such as styrene elastomer, olefin elastomer, vinyl chloride elastomer, urethane elastomer, polyester elastomer, polyamide elastomer, fluorine elastomer, 1,2-polybutadiene, or trans 1,4-polyisoprene; and polymer alloy, for example, PC resin/St resin alloy such as PC/ABS; PVC resin/St resin alloy such as PVC/ABS; PA resin/St resin alloy such as PA/ABS; PA resin/TPE alloy; PA resin/polyolefin resin alloy such as PA/PP; PC resin/PEs resin alloy such as PBT resin/TPE or PC/PBT; alloy of olefin resins, such as polyolefin resin/TPE or PP/PE; PPE resin alloy such as PVC/PMMA.

[0044] If the impact strength modifier is incorporated into these resins, known stabilizers and fillers can be added during the compounding, kneading or molding according to the intended use, as far as the physical properties of these resins are not impaired.

[0045] The impact strength modifier is preferably incorporated in an amount within a range from 0.01 to 50 parts, more preferably from 0.1 to 30 parts by weight, and particularly preferably from 1 to 20 parts by weight, based on 100 parts by weight of these resins.

[0046] Examples of the stabilizer include metal-based stabilizers, for example, lead stabilizer such as tribasic lead sulfate, dibasic lead phosphite, basic lead sulfite, or lead silicate; metal soap stabilizer derived from a metal such as potassium, magnesium, barium, zinc, cadmium, or lead and fatty acid such as 2-ethylhexanoic acid, lauric acid, myristic acid, palmitic acid, stearic acid, isostearic acid, hydroxystearic acid, oleic acid, ricinoleic acid, linoleic acid, or behenic acid; organotin stabilizer derived from an alkyl group or an ester group and fatty acid salt, maleate or sulfide; composite metal soap stabilizer such as Ba-Zn, Ca-Zn, Ba-Ca-Sn, Ca-Mg-Sn, Ca-Zn-Sn, Pb-Sn, or Pb-Ba-Ca; metal salt stabilizer derived usually from a group of metal such as barium or zinc and two or more organic acids such as a branched fatty acid (e.g. 2-ethylhexanoic acid, isodecanoic acid, trialkylacetic acid, etc.), unsaturated acid (e.g. oleic acid, ricinoleic acid, linoleic acid, etc.), alicyclic acid (e.g. naphthenic acid, etc.) and aromatic acid (e.g. carbolic acid, benzoic acid, salicylic acid, substituted derivatives thereof, etc.); and a metal salt liquid stabilizer obtained by dissolving these stabilizers in an organic solvent such as petroleum-based hydrocarbon, alcohol or glycerin derivative, and incorporating auxiliary stabilizers such as phosphite, epoxy compounds, color development inhibitors, transparency modifiers, photostabilizers, antioxidants, plate-out inhibitors and lubrincants; and

non-metal-based stabilizers, for example, epoxydated compound such as epoxy resin, epoxidated soybean oil, epox-

idated vegetable oil, or epoxidated fatty acid alkyl ester; organophosphite which contains phosphorous substituted with an alkyl group, an aryl group, a cycloalkyl group, or an alkoxy group and also contains a dihydric alcohol (e.g. propylene glycol, etc.) or an aromatic compound (e.g. hydroquinone, bisphenol A, etc.); hindered phenol such as bisphenol dimerized by BHT, sulfur, or a methylene group; ultraviolet absorber such as salicylate, benzophenone, or benzotriazole; photostabilizer such as hindered amine or nickel complex salt; ultraviolet screening agent such as carbon black or Rutile type titanium oxide; polyhydric alcohol such as trimethylolpropane, pentaerythritol, sorbitol, or mannitol; nitrogencontaining compound such as  $\beta$ -aminocrotonate, 2-phenylindole, diphenylthiourea, or dicyandiamide; sulfur-containing compound such as dialkylthiodipropionate; keto compound such as acetoacetate, dehydroacetic acid, or  $\beta$ -diketone; organosilicon compound; and borate.

[0047] Examples of the filler include carbonate such as ground limestone, precipitated limestone, or gelatinous limestone; inorganic fillers such as titanium oxide, clay, talc, mica, silica, carbon black, graphite, glass beads, glass fiber, carbon fiber, and metal fiber; organic fillers such as organic fiber made of polyamide, and silicone; and natural organic material such as wood flour.

[0048] There can also be added impact strength modifiers such as MBS, ABS, AES, NBR, EVA, chlorinated polyethylene, acrylic rubber, polyalkyl (meth)acrylate rubber graft copolymer, and thermoplastic elastomer; processing aids such as (meth)acrylate copolymer; alkyl esters of aromatic polybasic acids, such as dibutyl phthalate, dioctyl phthalate, diisodecyl phthalate, diisononyl phthalate, diundecyl phthalate, trioctyl trimellitate, triisooctyl trimellitate, and pyromellitate; alkyl esters of aliphatic polybasic fatty acids, such as dibutyl adipate, dioctyl adipate, dithiononyl adipate, dibutyl azelate, dioctyl azelate, and diisononyl azelate; phosphates such as tricresyl phosphate; polyester plasticizers such as those obtained by sealing the ends of a polycondensate having a molecular weight within a range from about 600 to 8,000, which is prepared from a polyhydric carboxylic acid (e.g. adipic acid, azelaic acid, sebacic acid, phthalic acid, etc.) and a polyhydric alcohol (e.g. ethylene glycol, 1,2-propylene glycol, 1,2-butylene glycol, 1,3-butylene glycol, 1,4-butylene glycol, etc.), with a monohydric alcohol or a monohydric carboxylic acid; epoxy plasticizers such as epoxidated soybean oil, epoxidated linseed oil, and epoxidated tall oil fatty acid-2-ethylhexyl; plasticizers such as chlorinated paraffin. There can also be added lubricant. Examples of the lubricant include pure hydrocarbons such as liquid paraffin and low-molecular polyethylene; halogenated hydrocarbon; fatty acids such as higher fatty acid and oxyfatty acid; fatty amides such as fatty amide; polyhydric alcohol esters of fatty acids, such as gyceride; fatty alcohol ester (ester wax) of fatty acid; metal soap; fatty alcohol; polyhydric alcohol;, polyglycol; polyglycerol; esters such as partial ester of fatty acid and polyhydric alcohol, and partial ester of fatty acid and polyglycol or polyglycerol; (meth) acrylate copolymer.

[0049] There can also be added flame retardants such as chlorinated paraffin, aluminum hydroxide, antimony trioxide, and halogen compound; heat resistance modifiers such as (meth)acrylate copolymer, imide copolymer, and styrene-acrylonitrile copolymer; mold release agents; nucleating agents; fluidity modifiers; colorants; antistatic agents; conductivity-imparting agents; surfactants; anti-fogging agents; blowing agents; and anti-fungus agents.

[0050] A method of preparing a resin composition by incorporating an impact strength modifier into a resin is not specifically limited, but a melt-mixing method is particularly preferred. A small amount of a solvent may be used, but not be used usually.

[0051] Examples of the mixing device include an extruder, Banbury mixer, roller, and kneader, and these mixing devices are operated in a batch-wise manner or continuously. There is no limitation on the mixing order.

### 40 EXAMPLES

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[0052] The following Examples and Comparative Examples further illustrate the present invention in detail. In the description, "parts" and "percentages" are by weight unless otherwise stated.

(Preparation Example 1) Preparation of acrylic rubber graft copolymer (1)

[0053] 99.5 Parts of 2-ethylhexyl acrylate and 0.5 parts of allyl methacrylate were mixed to obtain 100 parts of a (meth)acrylate monomer mixture. To 195 parts of distilled water, in which 1 part of dipotassium alkenylsuccinate was dissolved, 100 parts of the (meth)acrylate monomer mixture was added. After preliminarily stirring at 10,000 rpm using a homomixer, the mixture was emulsified and then dispersed under a pressure of 300 kg/cm² by using a homogenizer to obtain a (meth)acrylate emulsion.

[0054] This mixed solution was transferred to a separable flask equipped with a condenser and a stirring blade and then heated while replacing by nitrogen and mixing with stirring. Upon reaching 50°C, 0.5 parts of tert-butyl hydroper-oxide was added and, after heating to 50°C, a mixed solution of 0.002 parts of ferrous sulfate, 0.006 parts of disodium ethylenediamine tetraacetate, 0.26 parts of Rongalit and 5 parts of distilled water was added. Then, the polymerization was completed by being left to stand for five hours to obtain an acrylic rubber latex (A1).

[0055] The polymerization rate of the resulting acrylic rubber latex (A1) was 99.9%. This latex was solidified with ethanol and then dried to obtain a solid and the gel content was measured by extracting with toluene at 90°C for 12

hours. As a result, the gel content was 92.4%.

[0056] The acrylic rubber latex (A1) was charged in a separable flask equipped with a stirrer. The solid content of poly 2-ethylhexyl acrylate and allyl acrylate in the system was 10 parts. After adding distilled water to make the amount of distilled water in the system 195 parts, a mixed solution of 78 parts of n-butyl acrylate containing 2.0% allyl methacrylate, which constitutes the acrylic rubber (A2) component, and 0.32 parts of tert-butyl hydroperoxide was charged, and then this mixed solution was made to penetrate into the acrylic rubber latex (A1) particles by stirring for 10 minutes. Furthermore, 0.5 parts of polyoxyethylene ether sulfate was added and, after stirring for 10 minutes, the system was replaced by nitrogen and heated to 50°C. A mixed solution of 0.002 parts of ferrous sulfate, 0.006 parts of disodium ethylenediamine tetraacetate, 0.26 parts of Rongalit and 5 parts of distilled water was charged and the radical polymerization was initiated. Then, the polymerization was completed by maintaining at an inner temperature of 70°C for two hours to obtain a latex of a polyalkyl (meth)acrylate rubber (A).

[0057] A portion of this latex was collected and the particle size distribution of the polyalkyl (meth)acrylate rubber (A) was measured. This latex was dried to obtain a solid and the gel content was measured by extracting with toluene at 90°C for 12 hours. As a result, the gel content was 98.3%.

[0058] To the latex of the polyalkyl (meth)acrylate rubber (A), a mixed solution of 0.06 parts of tert-butyl hydroperx-oxide and 12 parts of methyl methacrylate was added dropwise at 70°C over 15 minutes, and then the graft polymerization of the vinyl monomer (B) to the polyalkyl (meth)acrylate rubber (A) was completed by maintaining at 70°C for four hours. The polymerization rate of the methyl methacrylate was 99.4%.

[0059] The resulting acrylic rubber graft copolymer latex was added dropwise in 200 parts of hot water containing 1.5% calcium chloride, followed by solidification, separation, washing and further drying at 75°C for 16 hours to prepare a powdered acrylic rubber graft copolymer (1), which was used as an impact strength modifier (1).

[0060] The composition is shown in Table 1.

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					Table 1	-			
		}	Polyal	Polyalkyl (meth)acrylate rubber (A)	th) acry	late ru	bber (A		Acrylic rubber
Preparation									graft copolymer
Example No.		Al COR	Al component		A2	component	ent	A1 component/ A2	Graft ratio
	2EHA	SMA	BA	AMA	BA	2EHA	AMA	component	(A)/(B)
	(%)	( <del>&amp;</del> )	(%)	( <del>%</del> )	(%)	(%)	(&)	(weight ratio)	(ueight ratio)
1	99.5		•	0.5	86		2	10/78	ישכילוור דמרזה
7	99.5			0.5	86		,	20/60	21/00
3	99.5			2	80		3	99/07	88/12
					30		7	30/58	88/12
7	99.0			1.0	98	•	7	30/55	88/12
2	99.0	-	•	1.0	86		2.	10/75	85/15
9	99.0	•		1.0	98		2	75/10	01/10
7	99.0	-		-	a v	0 2	1	27/00	CT/C0
æ	000			,		3	7	30/55	85/15
0	7007	•	•		98	•	2	30/55	85/15
'n	70.0	29	•	1.0	86		2	30/55	85/15
10	70.0		29	1.0	98		2	30/55	04/50
11		-			86	,	,	56/05	61/58
12	63.0		35	0			*	69/0	85/15
13				0 9				85/0	85/15
77		•	•	•	0	_	•		

[0061] In Table 1, a graft ratio is a weight ratio in case graft polymerization of the polyalkyl (meth) acrylate rubber (A) and the vinyl monomer (B) is conducted.

The abbreviations in Table 1 are as follows.

2EHA: 2-ethylhexyl acrylate SMA: stearyl methacrylate BA: butyl acrylate

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AMA: allyl methacrylate MMA: methyl methacrylate

[0062] The glass transition temperature of the acrylic rubber graft copolymer (1) was measured. This measured glass transition temperature is a value derived from the polyalkyl (meth) acrylate rubber (A). The results are shown in Table 2. [0063] The particle size distribution and glass transition temperature were measured by the following procedure.

1) Measurement of particle size distribution of polyalkyl (meth)acrylate rubber (A)

Using samples prepared by diluting the resulting latex with distilled water, the particle size distribution was measured by using a particle size distribution meter, Model CHDF2000, manufactured by MATEC USA Co. The measurement was conducted under the standard conditions recommended by MATEC Co. Using an exclusive capillary type cartridge for particle separation and a carrier solution, 0.1 ml of each diluted latex sample having a concentration of about 3% was used for measurement under the fixed conditions of neutral pH, a flow rate of 1.4 ml/minute, a pressure of about 4000 psi and a temperature of 35°C. As the particle size standard, 12 kinds of monodisperse polystyrene having a known particle size within a range from 0.02 to 0.8  $\mu$ m (manufactured by DUKE USA Co.) were used.

2) Measurement of glass transition temperature of acrylic rubber graft copolymer

A plate, which was obtained by forming 70 parts of the resulting acrylic rubber graft copolymer (1) and 30 parts of methyl polymethacrylate (PMMA) into pellets using a 25 $\phi$  single-screw extruder at 250°C and forming pellets into a plate having a thickness of 3 mm using a press machine set at 200°C, was cut into samples having a width of 10 mm and a length of 12 mm. Using a meter, Model DMA983, manufactured by TA Instruments Co., the measurement was conducted under the conditions of a heating rate of 2°C/minute and the temperature corresponding to the transition point of the resulting Tan  $\delta$  curve was determined as the glass transition temperature.

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Table 2

(Preparation Examples 2 to 10) Preparation of acrylic rubber graft copolymers (2) to (10)

[0064] Using the same procedure as in Preparation Example 1, except that the monomers which constitute the acrylic rubber (A1) component and the acrylic rubber (A2) component, and the weight ratio thereof as well as the graft ratio of the polyalkyl (meth) acrylate rubber (A) to the vinyl monomer (B) were as shown in Table 1, acrylic rubber graft copolymers (2) to (10) were prepared and used as impact strength modifiers (2) to (10).

[0065] Using the same procedure as in Preparation Example 1, the particle size distribution of the polyalkyl (meth) acrylate rubber (A) and the glass transition temperature of the acrylic rubber graft copolymers (2) to (10) were measured. The results are shown in Table 2.

[0066] The particle size distribution of the polyalkyl (meth)acrylate rubber (A) prepared in Preparation Example 2 is shown in Fig. 1 and Fig. 2. Fig. 1 shows a number distribution, while Fig. 2 shows a weight distribution.

(Preparation Example 11) Preparation of acrylic rubber graft copolymer (11)

[0067] 295 Parts of distilled water and 0.4 parts of sodium dodecylbenzenesulfonate were charged in a separable flask equipped with a stirrer and heated to 50°C after replacing by nitrogen. After charging a mixed solution of 85 parts of n-butyl acrylate containing 2% allylmethacrylate and 0.4 parts of tert-butyl hydroperoxide and stirring for 30 minutes, a mixed solution of 0.002 parts of ferrous sulfate, 0.006 parts of disodium ethylenediamine tetraacetate, 0.26 parts of Rongalit and 5 parts of distilled water was charged and the radical polymerization was initiated, and then the polymerization was completed by maintaining at an inner temperature of 70°C for two hours to obtain a polyalkyl (meth) acrylate rubber latex. A portion of this latex was collected and the average particle size of the polyalkyl (meth) acrylate rubber was measured. The result was 0.22 μm and the particle size distribution had only one peak. This latex was dried to obtain a solid and the gel content was measured by extracting with toluene at 90°C for 12 hours. The result was 97.3%.

25 [0068] To this polyalkyl (meth)acrylate rubber latex, a mixed solution of 0.06 parts of tert-butyl hydroperxoxide and 15 parts of methyl methacrylate was added dropwise at 70°C over 15 minutes, and then the graft polymerization to the polyalkyl (meth) acrylate rubber was completed by maintaining at 70°C for four hours.

[0069] The polymerization rate of the methyl methacrylate was 97.2%.

[0070] The resulting graft copolymer latex was added dropwise in 200 parts of hot water containing 1.5% calcium chloride, followed by solidification, separation, washing and further drying at 75°C for 16 hours to prepare a powdered acrylic rubber graft copolymer (11), which was used as an impact strength modifier (11). The composition is shown in Table 1. Using the same procedure as in Preparation Example 1, the glass transition temperature of the acrylic rubber graft copolymer (11) was measured. The results are shown in Table 3.

35 (Preparation Example 12) Preparation of acrylic rubber graft copolymer (12)

[0071] The acrylic rubber (A1) component prepared according to the composition shown in Table 1 was subjected to graft polymerization as it is without adding the acrylic rubber (A2) component thereto. The polymerization rate of the methyl methacrylate was 98.1%. The resulting graft copolymer latex was added dropwise in 200 parts of hot water containing 1.5% calcium chloride, followed by solidification, separation, washing and further drying at 75°C for 16 hours to prepare a powdered acrylic rubber graft copolymer (12), which was used as an impact strength modifier (12).

[0072] The composition is shown in Table 1. Using the same procedure as in Preparation Example 1, the particle size distribution of the polyalkyl (meth) acrylate rubber and the glass transition temperature of the acrylic rubber graft copolymer (12) were measured. The results are shown in Table 3.

(Preparation Example 13) Preparation of acrylic rubber graft copolymer (13)

[0073] Using the same procedure as in Preparation Example 11, except that the amount of sodium dodecylbenzenesulfonate was changed to 1.0 parts, an acrylic rubber graft copolymer latex, with a particle size distribution having only one peak located at 0.10  $\mu$ m, was obtained. In the same procedure as in Preparation Example 11, solidification, separation, washing and further drying were conducted to prepare a powdered acrylic rubber graft copolymer (13), which was used as an impact strength modifier (13).

[0074] The composition is shown in Table 1. Using the same procedure as in Preparation Example 1, the glass transition temperature of the acrylic rubber graft copolymer (13) was measured. The results are shown in Table 3.

(Preparation Example 14) Preparation of acrylic rubber graft copolymer (14)

[0075] After the acrylic rubber graft copolymer latex having a particle size of 0.22 µm obtained in Preparation Example

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11 was mixed with the acrylic rubber graft copolymer latex obtained in Preparation Example 13 so that {number  $(N_{0.1})$  of particles having a center particle size of 0.1  $\mu$ m} / {number  $(N_{0.2})$  of particles having a center particle size of 0.22  $\mu$ m} becomes 2, the mixture was added dropwise in 200 parts of hot water containing 1.5% calcium chloride, followed by solidification, separation, washing and further drying at 75°C for 16 hours to prepare a powdered acrylic rubber graft copolymer (14), which was used as an impact strength modifier (14).

[0076] Using the same procedure as in Preparation Example 1, the particle size distribution of the polyalkyl (meth) acrylate rubber and the glass transition temperature of the acrylic rubber graft copolymer (14) were measured. The results are shown in Table 3.

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			Table 3	3		
	Impact streng	Impact strength modifier incorporated	corporated	Particle si	size distribution of proper (meth) acrylate rubber	Particle size distribution of polyalkyl (meth) acrylate rubber
Comp. Example	Type	Glass tr temperat	Glass transition temperature (C)	1 1 1 1 1 1 1 1 1 1 1 1 1 1 1 1 1 1 1 1	Peak	Proportion of particles having a
0 2	strength modifier	derived from Al component	derived from derived from Al component A2 component	state	position (µm)	particle size ranging from 0.05 to 0.4 $\mu$ m (% by weight)
1	(11)	•	- 28	Mono dispersion	0.22	
7	(12)	- 44	•	Mono dispersion	0.24	
æ	(13)	,	- 28	Mono dispersion	0.10	. 06 <
4	(14)	,	-28	Double dispersion	0.22	06 <
S	Metablen C-223		-61	,	07.0	

(Examples 1 to 10)

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[0077] The impact resistance and weathering resistance of each of the vinyl chloride resin compositions obtained by using the acrylic rubber graft copolymers (1) to (10) as the impact strength modifiers (1) to (10), respectively, and incorporating them into rigid vinyl chloride resins, were evaluated by the following procedure.

[0078] The results are shown in Table 4.

1) Impact resistance: evaluated by Izod impact resistance. Vinyl chloride compositions were prepared according to the following two kinds of compositions (compositions A and B) and extruded into square bars of 1/2 inch  $\times$  1/4 inch in size using a 25 mm $\phi$  single-screw extruder controlled to  $190^{\circ}$ C, and then the impact resistance was evaluated by the method defined in ASTM D256. The Izod impact resistance of the vinyl chloride resin composition of composition A was measured at  $23^{\circ}$ C, while that of the vinyl chloride resin composition of composition B was measured at  $0^{\circ}$ C, which are respectively described as measurement A and measurement B in the table.

Composition A	Vinyl chloride resin (polymerization degree: 700)	100 parts	l
	Dibutyltin maleate	3.5 parts	
	Stearl alcohol	0.8 parts	
	Processing aid (Metablen P-700)	0.4 parts	
	Carbon black	0.5 parts	
	Modifier	7.5 parts	1
Composition B	Vinyl chloride resin (polymerization degree: 1100)	100 parts	
	Dibasic lead phosphite	2.5 parts	
	Dibasic lead stearate	0.7 parts	
	Lead stearate	0.5 parts	
	Calcium stearate	0.9 parts	
	Polyethylene wax (molecular weight: 2200)	0.1 parts	
	Calcium carbonate	5.0 parts	
	Processing aid (Metablen P-501)	1.0 parts	
	Carbon black	0.5 parts	
	Modifier	7.5 parts	

2) Weathering resistance: An operation of subjecting each of the sheet specimens obtained by preparing vinyl chloride resin compositions according to the composition B, kneading at 200°C for 6 minutes using a 6 inch test roll machine manufactured by Kansai Roll Co., Ltd. at 15 rpm for five minutes, interposed between dies at 200°C and cooling while applying a pressure of 50 kg/cm², to an Eyesuper UV tester controlled to a temperature of 60°C, manufactured by Dainippon Plastics Co., Ltd., for eight hours and leaving to stand in a thermo-hygrostat controlled to a temperature of 60°C and a humidity of 95% for 16 hours was repeated five times, and then the appearance of each specimen sheet was visually evaluated.

[0079] In Table 4, the symbol O indicates "good", while the symbol × indicates "poor".

(Comparative Examples 1 to 5)

[0080] In the same procedure as in Example 1, except that the acrylic rubber graft copolymers (11) to (14) were used as the impact strength modifiers (11) to (14) in Comparative Examples 1 to 4 and a commercially available MBS modifier "Metablen C-223" manufactured by Mitsubishi Rayon was used as the impact strength modifier in Comparative Example 5, and then incorporated into the vinyl chloride resin, vinyl chloride resin compositions were prepared.

[0081] The impact resistance and weathering resistance of the resulting vinyl chloride resin compositions were evaluated using the same procedure as in Example 1.

[0082] The results are shown in Table 4.

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CHOOL CLUMENT	The same of the same	Physical properties of vinyl chloride resin composition	inyl chloride resi	in composition
Evample and Comp.	strength modifier	Weathering resistance	Impact resistance (J/m)	tance (J/m)
		Fade resistance	Measurement A	Measurement B
Example 1	(1)	0	1090	190
Example 2	(2)	0	950	210
Example 3	(3)	Ο	880	300
Example 4	(4)	Ο	1090	350
Example 5	(5)	0	850	210
Example 6	(9)	0	006	330
Example 7	(7)		1220	400
Example 8	(8)	0	1180	480
Example 9	(6)	0	720	180
Example 10	(10)	0	780	190
Comp. Example 1	(11)	0	110	3.0
Comp. Example 2	(12)	0	140	09
Comp. Example 3	(13)	0	130	30
Comp. Example 4	(14)	0	110	0.9
Comp. Example 5	Metablen C-223	×	1120	430

Table 4

[0083] As shown in Table 4, the vinyl chloride resin compositions of Examples 1 to 10 incorporated with the acrylic rubber graft copolymers (1) to (10) of Preparation Examples 1 to 10 as the impact strength modifier were superior in both the weathering resistance and impact resistance and also exhibited good impact resistance at low temperatures. [0084] As shown in Fig. 1, the number distribution of the particle size of the polyalkyl (meth)acrylate rubber of Preparation Example 2 was a double dispersion distribution having at least two peaks at 0.11  $\mu$ m and 0.19  $\mu$ m. As shown in Fig. 2, not less than 90% by weight of the total weight of the polyalkyl (meth)acrylate rubber consists of particles having a particle size within a range from 0.05 to 0.4  $\mu$ m. The particle size distribution of each of polyalkyl (meth) acrylate rubbers of Preparation Examples 1 and 3-10 exhibited almost the same double distribution, although the graphs are omitted.

[0085] As shown in Table 2, any of the acrylic rubber graft copolymers (1) to (10) of Preparation Examples 1 to 10 had a glass transition temperature (Tgl) derived from the acrylic rubber (A1) component and a glass transition temperature (Tg2) derived from the acrylic rubber (A2) component and, furthermore, Tgl was lower than Tg2 and both are lower than 10°C.

[0086] On the other hand, regarding the acrylic rubber graft copolymers (11) to (14) of Preparation Examples 11 to 14, the polyalkyl (meth)acrylate rubber (A) is not composed of the acrylic rubber (A1) component and the acrylic rubber (A2) component. Therefore, the vinyl chloride resin compositions of Comparative Examples 1 to 4 incorporating the acrylic rubber graft copolymers (11) to (14) were inferior in impact resistance. The vinyl chloride resin composition of Comparative Example 5 incorporating the MBS modifier "Metablen C-223" was superior in impact resistance, but exhibited poor weathering resistance because fading occurred in the specimen after subjecting to the weathering test.

[0087] Then, the impact strength modifiers (2, 5) obtained in the Preparation Examples described above and a new impact strength modifier (15) prepared by changing the vinyl monomer (B) component were mixed with the thermoplastic resin other than the vinyl chloride resin, and the impact strength of each was measured (Examples 11 to 15).

(Preparation Example 15) Preparation of acrylic rubber graft copolymer (15)

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[0088] Using the same procedure as in Preparation Example 2, except that the vinyl monomer (B) component was replaced by a copolymer component wherein the ratio of the weight  $(M_{AN})$  of acrylonitrile to the weight  $(M_{ST})$  of styrene,  $M_{AN}$ : $M_{ST}$ , was 1:3, and that ratio is as shown in Table 5, the solidification, separation, washing and further drying were conducted to prepare a powdered acrylic rubber graft copolymer (15), which was used as an impact strength modifier (15).

[0089] In the same procedure as in Preparation Example 1, the particle size distribution of the polyalkyl (meth) acrylate rubber and the glass transition temperature of the acrylic rubber graft copolymer (15) were measured. The results are shown in Table 6.

35 (Examples 11 to 15)

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[0090] After each of the impact strength modifiers (2, 5, 15) was mixed with a thermoplastic resin according to each composition shown in Table 7 using a Henschel mixer for four minutes, the mixture was melt-kneaded at a cylinder temperature of 260°C using a 30 mm¢ twin-screw extruder and then formed into pellets made of each of the thermoplastic resin compositions. These pellets were injection-molded to obtain 1/4 inch Izod specimens. The impact strength of each specimen was measured. The results are shown in Table 8. A molded article made of any of the resin compositions exhibited good appearance. As the thermoplastic resin, the following resins were used.

Polycarbonate resin: Bisphenol A type polycarbonate (hereinafter referred to as PC) having a viscosity-average molecular weight of about 22000 was used.

Polyester resin: Polytetramethylene terephthalate (hereinafter referred to as PBT) having an intrinsic viscosity [η] of 1.05 was used.

Polyphenylene ether resin: Poly(2,6-dimethyl-1,4-phenylene)ether (manufactured by Japan GE Plastics Co., Ltd.) (hereinafter referred to as PPE) having an intrinsic viscosity (chloroform, 25°C) of 0.48 dl/g was used.

Polystyrene resin: Tobolex 876-HF (high-impact polystyrene (hereinafter referred to as HIPS), manufactured by Mitsui Chemicals Co., Ltd.) was used.

ABS resin: Dia Pet 3001 manufactured by Mitsubishi Rayon (hereinafter referred to as ABS) was used.

SAN resin: SR30B manufactured by Ube Saikon Co., Ltd. (hereinafter referred to as SAN) was used.

PPS resin: Torprene PPS manufactured by Tonen Chemicals Co., Ltd. (hereinafter referred to as PPS) was used.

Table 5

Preparation         A1 component         A2 component         A1 component           Example No.         2EHA SMA BA AMA BA 2EHA AMA A2 component         A2 component           (%) <th></th> <th></th> <th>Polyalk</th> <th>Polyalkyl (meth) acrylate rubber (A)</th> <th>h) acryl</th> <th>ite rubk</th> <th>oer (A)</th> <th></th> <th>Acrylic rubber</th> <th>Acrylic rubber graft copolymer</th>			Polyalk	Polyalkyl (meth) acrylate rubber (A)	h) acryl	ite rubk	oer (A)		Acrylic rubber	Acrylic rubber graft copolymer
2EHA SMA BA AMA BA 2EHA AMA (%) (%) (%) (%) (%) (%) (%) (%) (%) (%)	Preparation		A1 com	ponent		A2	compone	nt	A1 component/	Graft ratio
(%) (%) (%) (%) (%) (%) (%) (%) (%) (%)	Example No.	2ЕНА	SMA	ВА	AMA	BA	2EHA	AMA	A2 component	(A) / (B)
- 2		(%)	(&)	(%)	(%)	(%)	(%)	(%)	(weight ratio)	(weight ratio)
	15	3.66	-		5.0	86	•	7	10/78	82/18

Pable 6

	Impact	act strength modifier	odifier	Particle s (me	ize distrik th)acrylate	Particle size distribution of polyalkyl (meth)acrylate rubber (A)
Preparation Example No.	Types of impact	Glass transition temperature (C)	ansition ure (C)	Distribution	Peak	Proportion of particles having a particle
	strength modifier	derived from derived from Al component A2 component	derived from A2 component	state	(m m)	size ranging from 0.05 to 0.4 $\mu$ m (% by weight)
15	(15)	-55	-28	Double distribution	0.20	06 <
					0.10	

Table 7

	$\overline{}$	Т	Τ	1	Т	Т
odifier	(15)			3	•	
Impact strength modifier incorporated and parts	(5)	•				1.5
Impact	(2)	15	3		3	
rts	PPS					95
nermoplastic resin incorporated and parts	HIPS			09		
porated	PPE	٠		40	-	•
in inco	SAN	•	•	,	13	,
tic res	ABS	•	٠	•	10	•
rmoplas	PBT	20	•	•	•	•
The	PC	35	9.7	•	74	,
Example No.		1.1	1.2	13	14	15

Table 8

Example No.	Izod impact strength (J/m)
11	770
12	600
13	210
14	700
15	210

[0091] As is apparent from the results shown in Table 8, the impact strength modifier of the present invention can improve the impact resistance of any of the thermoplastic resins shown in the Examples.

### INDUSTRIAL APLICABILITY

[0092] As described above, since this impact strength modifier comprises, as a principal component, an acrylic rubber graft copolymer which is obtained by graft polymerization of a polyalkyl (meth)acrylate (A) rubber containing a plurality of acrylic rubber components composed of a specific monomer as a principal component and a vinyl monomer (B), it is made possible to improve the impact resistance, particularly the impact resistance at low temperatures, of the resin and to satisfactorily maintain the weathering resistance by adding a small amount of the impact strength modifier to the resin. Accordingly, a molded article of a resin composition incorporating this impact strength modifier is superior in impact resistance and also satisfactorily maintains the weathering resistance such as fade resistance or impact resistance. This impact strength modifier is suited for incorporation into a thermoplastic resin, particularly a vinyl chloride resin.

### Claims

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- 1. An impact strength modifier comprising, as a principal component, an acrylic rubber graft copolymer which is obtained by graft polymerization of a polyalkyl (meth)acrylate rubber (A) with one or more vinyl monomers (B), said polyalkyl (meth)acrylate rubber (A) comprising, as a principal component, an acrylic rubber (A1) component containing at least one of a (meth)acrylate of an alcohol having a branched side chain or an alcohol having an alkyl group with 13 or more carbon atoms and a (meth)acrylate having a hydroxyl group, a methoxy group or an ethoxy group as a constituent component and an acrylic rubber (A2) component containing n-butyl acrylate as a constituent component.
- 2. An impact strength modifier comprising, as a principal component, an acrylic rubber graft copolymer which is obtained by graft polymerization of a polyalkyl (meth) acrylate rubber (A) with one or more vinyl monomers (B), said polyalkyl (meth)acrylate rubber (A) comprising, as a principal component, an acrylic rubber (A1) component containing at least one of a (meth)acrylate of an alcohol having a branched side chain or an alcohol having an alkyl group with 13 or more carbon atoms as a constituent component and an acrylic rubber (A2) component containing n-butyl acrylate as a constituent component.
- An impact strength modifier according to claim 1, wherein said polyalkyl (meth)acrylate rubber (A) has two or more glass transition temperatures at 10°C or lower.
- 4. An impact strength modifier according to claim 1, wherein a glass transition temperature (TgI) derived from said acrylic rubber (A1) component is lower than a glass transition temperature (Tg2) derived from said acrylic rubber (A2) component.
  - 5. An impact strength modifier according to claim 1, wherein said polyalkyl (meth)acrylate rubber (A) comprises said acrylic rubber (A1) component containing at least one of 2-ethylhexyl acrylate, ethoxyethyl acrylate, methoxypropylene glycol acrylate, 4-hydroxybutyl acrylate, tridecyl methacrylate and stearyl methacrylate as a constituent component and said acrylic rubber (A2) component containing n-butyl acrylate as a constituent component, and wherein a glass transition temperature (Tgl) derived from said acrylic rubber (A1) component is lower than a glass transition temperature (Tg2) derived from said acrylic rubber (A2) component.

- 6. An impact strength modifier according to claim 1, wherein said polyalkyl (meth)acrylate rubber (A) comprises said acrylic rubber (A1) component containing at least one of 2-ethylhexyl acrylate, tridecyl methacrylate and stearyl methacrylate as a constituent component and said acrylic rubber (A2) component containing n-butyl acrylate as a constituent component.
- An impact strength modifier according to claim 1, wherein said polyalkyl (meth) acrylate rubber (A) contains 5-95% by weight of said acrylic rubber (A1) component and 95-5% by weight of said acrylic rubber (A2) component.
- 8. An impact strength modifier according to claim 1, wherein said polyalkyl (meth)acrylate rubber (A) contains a monomer having two or more unsaturated bonds per molecule in an amount of 20% by weight or less.

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- 9. An impact strength modifier according to claim 1, wherein said acrylic rubber (A1) component and said acrylic rubber (A2) component respectively contain a monomer having two or more unsaturated bonds in a molecule in an amount within a range from of 0.1 to 20% by weight.
- 10. An impact strength modifier according to claim 1, wherein the number distribution of the particle size of said polyalkyl (meth)acrylate rubber (A) is a double dispersion distribution having at least two peaks within a range from 0.05 to 0.4 μm and not less than 70% by weight of said polyalkyl (meth)acrylate rubber (A) consists of particles having a particle size within a range from 0.05 to 0.4 μm.
- 11. An impact strength modifier according to claim 1, wherein said acrylic rubber graft copolymer comprises 5-95% by weight of said polyalkyl (meth)acrylate rubber (A) and 95-5% by weight of said vinyl monomer (B).
- 12. An impact strength modifier according to claim 1, wherein said acrylic rubber graft copolymer comprises 50-95% by weight of said polyalkyl (meth)acrylate rubber (A) and 50-5% by weight of said vinyl monomer (B).
  - 13. An impact strength modifier according to claim 1, wherein said acrylic rubber graft copolymer comprises 80-95% by weight of said polyalkyl (meth)acrylate rubber (A) and 20-5% by weight of said vinyl monomer (B).
- 30 14. An impact strength modifier according to claim 1, wherein said acrylic rubber graft copolymer comprises 82-95% by weight of said polyalkyl (meth)acrylate rubber (A) and 18-5% by weight of said vinyl monomer (B).
  - 15. An impact strength modifier according to claim 1, wherein said acrylic rubber graft copolymer is obtained by graft polymerization of said polyalkyl (meth)acrylate (A) rubber, which is obtained by polymerizing a monomer containing n-butyl acrylate constituting said acrylic rubber (A2) component in the presence of said acrylic rubber (A1) component obtained by emulsion polymerization, and one or more vinyl monomers (B).
  - 16. An impact strength modifier according to claim 1, wherein said acrylic rubber graft copolymer is obtained by graft polymerization of said polyalkyl (meth)acrylate (A) rubber, which is obtained by polymerizing a monomer containing n-butyl acrylate constituting said acrylic rubber (A2) component in the presence of said acrylic rubber (A1) component obtained by forced emulsion polymerization, and one or more vinyl monomers (B), and the number distribution of the particle size of said polyalkyl (meth)acrylate rubber (A) is a double dispersion distribution having at least two peaks within a range from 0.05 to 0.4 μm and not less than 70% by weight of said polyalkyl (meth)acrylate rubber (A) consists of particles having a particle size within a range from 0.05 to 0.4 μm.
  - 17. A method of preparing the impact strength modifier of claim 1, which comprises subjecting a monomer containing n-butyl acrylate constituting an acrylic rubber (A2) component to emulsion polymerization in the presence of an acrylic rubber (A1) component to obtain a polyalkyl (meth)acrylate (A) rubber (A).
- 18. A method of preparing the impact strength modifier according to claim 17, wherein said acrylic rubber (A1) component is obtained by a forced emulsion polymerization method.
  - 19. A thermoplastic resin composition comprising the impact strength modifier of claim 1 and a thermoplastic resin.
- 20. A vinyl chloride resin composition comprising the impact strength modifier of claim 1 and a rigid vinyl chloride resin.

FIG. 1

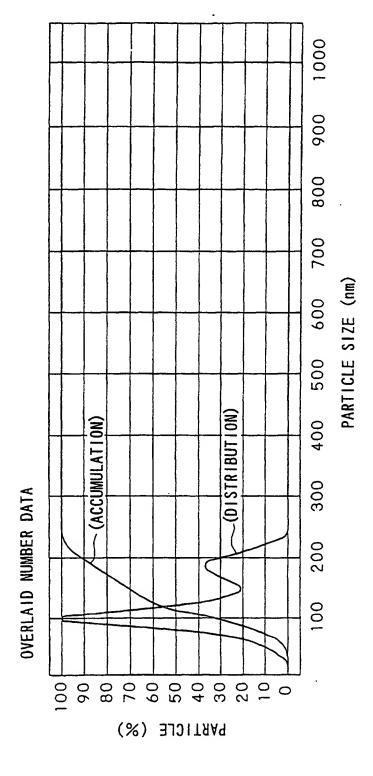
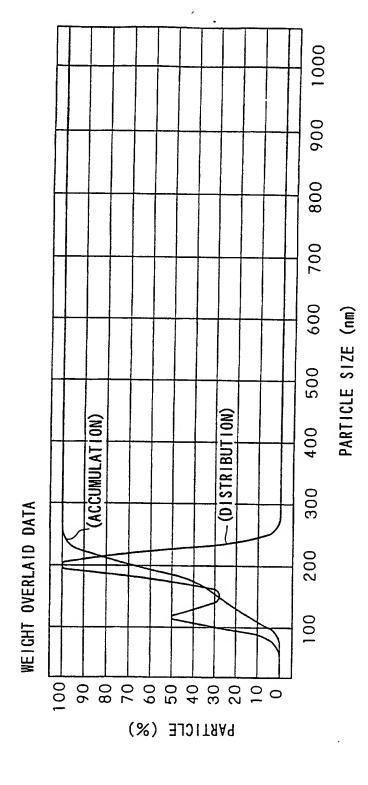


FIG 2



# INTERNATIONAL SEARCH REPORT

International application No.
PCT/JP99/03786

		<del></del>			
	SIFICATION OF SUBJECT MATTER .Clf C08L51/00, C08L27/06, C08	L101/00, C08F285/00	•		
According	to International Patent Classification (IPC) or to both n	ational classification and IPC			
<del></del>	S SEARCHED	<del></del>			
Minimum o Int	documentation searched (classification system followed CC1 CO8L51/00, CO8L27/06, CO8	d by classification symbols) L101/00, C08F251/00-29	2/00		
Documenta	tion searched other than minimum documentation to th	e extent that such documents are include	d in the fields searched		
Electronic	data base consulted during the international search (na	me of data base and, where practicable, so	earch terms used)		
C. DOCU	MENTS CONSIDERED TO BE RELEVANT				
Category*	Citation of document, with indication, where ap		Relevant to claim No.		
A	JP, 59-105012, A (Mitsubish 18 June, 1984 (18. 06. 84), Claims (Family: none)	i Chemical Corp.),	1-20		
<b>A</b>	JP, 59-108056, A (Mitsubish 22 June, 1984 (22. 06. 84), Claims (Family: none)	i Chemical Corp.),	1-20		
A	JP, 59-138257, A (Mitsubish: 8 August, 1984 (08. 08. 84), Claims (Family: none)		1-20		
A	JP, 5-247312, A (Rohm and Ham) 24 September, 1993 (24. 09. Claims; Par. No. [0012] & EP, 532234, Al & US, 519	93),	1-20		
Further documents are listed in the continuation of Box C. See patent family annex.					
'A' docume consider consider consider carlier cited docume cited to special 'O' docume means 'P' docume	categories of cited documents: ent defining the general state of the art which is not red to be of particular relevance document but published on or after the international filing date ent which may throw doubts on priority claim(s) or which is establish the publication date of another citation or other reason (as specified) ent referring to an oral disclosure, use, exhibition or other ent published prior to the international filing date but later than rity date claimed	"I" later document published after the interdate and not in conflict with the application the principle or theory underlying the interded and particular relevance; the character of the involve an inventive step to combined with one or more other such debeing obvious to a person skilled in the office of the same patent factor of the same patent factor."	tion but cited to understand vention aimed invention cannot be d to involve an inventive step aimed invention cannot be when the document is locuments, such combination att		
	actual completion of the international search eptember, 1999 (16. 09. 99)	Date of mailing of the international sear 28 September, 1999			
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### (54) IMPACT MODIFIERS FOR THERMOPLASTIC RESINS

(57) Excellent impact modifiers for vinyl chloride resins, which are copolymers prepared by polymerizing 10 to 70 wt% of a monomer component comprising 60 to 100 wt% of at least one vinyl monomer selected from the group consisting of(meth)acrylic esters, aromatic vinyl compounds, vinyl cyanide and vinyl chloride and 0 to 40 wt% of other copolymerizable monomer in the

presence of 30 to 90 wt% of hollow rubber particles which exhibit a glass transition temperature (Tg) of 0°C or below and a void content of 3 to 90 vol% as determined in the state of a latex and contain a crosslinking agent in an amount of 0.1 to 5 parts by weight per 100 parts by weight of the whole of the constituent monomers of the rubber.

### Description

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### **TECHNICAL FIELD**

5 [0001] The present invention relates to an impact modifier for a vinyl chloride resin.

### **BACKGROUND ART**

[0002] Conventionally, there have been various suggestions for improving impact resistance of a thermoplastic resin. For example, in case of vinyl chloride resins it was known that a copolymer of diene-based rubber or acrylate-based rubber is mixed thereto (Japanese Examined Patent Publication No. 39-19035/1964). Further, there are suggestions of a method of increasing a particle size of a rubber component (Japanese Examined Patent Publication No. 42-22541/1967) and a method of lowering Tg of a rubber component (Japanese Unexamined Patent Publication Nos. 2-1763/1990 and 8-100095/1996) to improve impact resistance. However, in these methods, there are problems that it is difficult to significantly improve impact resistance, and cost for raw material drastically increases.

[0003] For improving impact resistance of thermoplastic resins such as a vinyl chloride resin, stress concentration of a molded article, and formation and expansion of voids in a rubber play an important role. For obtaining stress concentration, it is essential to introduce a rubber component having elastic modulus significantly lower than that of a thermoplastic resin. Actually, optimization of a size and a shape of the rubber component has been investigated by introducing various rubbers. Formation and expansion of voids in the rubber are expected to contribute intensely to growth of shearing yield particularly having large energy absorption in an impact test, to lead to improvement in impact resistance of a thermoplastic resin which contains a rubber.

[0004] Therefore, the method to promote formation and expansion of voids in a rubber component at impact (under stress) of a molded article is extremely important, and formation and expansion of voids in the rubber component are considered to be significantly influenced by cross-linking state of the rubber. Also, when a rubber component is previously made hollow, there is possibility that voids are easily propagated under stress.

[0005] Then, relation among void (hollow) state of a rubber component in a latex, hollow state of a thermoplastic resin molded article mixed with an impact modifier containing a rubber component, and impact strength of the molded article has been investigated by changing an amount of a cross-linking agent controlling cross-linking condition of the rubber component. As a result, it has been found that an impact modifier comprising 30 to 90 % by weight of a hollow rubber of which Tg is at most 0 °C and an amount of a cross linking agent is 0 to 5 % by weight, and which is not hollow after molding and has a void ratio of 3 to 90 % by volume in a latex, and 10 to 70 % by weight of a monomer or monomers comprising 60 to 100 % by weight of at least one vinyl monomer selected from the group consisting of a (meth)acrylate compound, an aromatic vinyl compound and a vinyl cyanide compound, and 0 to 40 % by weight of a monomer copolymerizable therewith, has an extremely large effect on improving impact resistance of a thermoplastic resin. Namely, though it is basic to provide stress concentration by introducing a rubber component (Tg is at most 0 C°) having elastic modulus significantly lower than that of a thermoplastic resin such as a vinyl chloride resin which becomes a continuous phase of a molded article, it is also important to change the rubber particle to a void at impact (under stress) by decreasing an amount of a cross linking agent for this soft rubber component.

[0006] However, it has also been found that there is a tendency that a rubber particle collapses at molding and is finely dispersed and the rubber particle does not easily act as a stress concentration point when any cross linking agents become 0 % by weight. Further, it has been recognized from TEM observation and specific gravity measurement that a hollow rubber having a low content of a cross linking agent has voids filled with water in a latex, but in the processes of coagulation, thermal treatment, drying, blending and molding after graft polymerization, water in the rubber particle is removed, and after drying and molding, voids in the rubber collapse and disappear to become non-hollow. [0007] On the other hand, it has been found that voids in the rubber particle are easily formed and expanded at impact (under stress) in an impact modifier containing a hollow rubber having voids in the rubber in a latex to significantly improve in impact strength. A larger amount of the cross linking agent used in a rubber is more suitable to keep the rubber particle in a hollow state. But judging from easiness of total expansion of voids, the largest improving effect is obtained by using an impact modifier containing a rubber which is hollow in a latex and is not hollow at drying and after molding by controlling an amount of a cross linking agent in a rubber to a small amount thereof.

## **DISCLOSURE OF INVENTION**

[0:008] Namely, the present invention relates to an impact modifier for a vinyl chloride resin, which is prepared by polymerizing 10 to 70 % by weight of a monomer containing 60 to 100 % by weight of at least one vinyl monomer selected from the group consisting of a (meth)acrylate compound, an aromatic vinyl compound, a vinyl cyanide compound and vinyl chloride, and 0 to 40 % by weight of the other monomer copolymerizable therewith, in the presence

of 30 to 90 % by weight of a hollow rubber particle of which glass transition temperature Tg is at most  $0 \, ^{\circ}$ C, a void ratio is 3 to 90 % by volume in a latex state and which contains a cross linking agent in an amount of 0.1 to 5 parts by weight based on 100 parts by weight of the total rubber component monomer.

[0009] In the impact modifier, the amount of the cross linking agent is preferably 0.1 to 1.5 parts by weight.

[0010] In the impact modifier, the amount of the cross linking agent is more preferably 0.2 to 0.7 part by weight.

[0011] In the impact modifier, an average particle diameter of the hollow rubber in a latex state is preferably 50 to 300nm.

[0012] Further, the present invention relates to a vinyl chloride resin composition comprising a vinyl chloride resin containing at least 50 % by weight of vinyl chloride, and the above-mentioned impact modifier.

### BEST MODE FOR CARRYING OUT THE INVENTION

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[0013] There are various methods for preparing hollow rubber particle. For example, the following methods are well known ("Gosei Latex no Oyo (application of synthetic latex)", Takaaki Sugimura et al., p 285). The rubber particle of the present invention having voids in a latex may be prepared by any of these methods.

- (a) a method wherein a W/O/W emulsion is prepared, and a monomer in the O layer is polymerized (O: lipophilic, W: hydrophilic)
- (b) a method wherein core-shell particles having swellable core is swelled at a temperature of at least glass transition temperature of shell layer to make hollow
- (c) a method by using a two-step polymerization of polymers having different solubility parameter
- (d) a method wherein a polymerizable monomer containing a cross linkable monomer and a hydrophilic monomer, and an oil substance are finely dispersed in water to make an O/W emulsion, and the monomer is polymerized to remove the oily substance
- (e) a method by using migration of a carboxylic acid copolymerized into a particle under acidic or alkaline condition in the particle

[0014] There is no particular limitation for the hollow rubber component used in the present invention, as long. as it is a rubber elastomer of which glass transition temperature is at most  $0\,^{\circ}$ C. But the lower the glass transition temperature is, the better. The glass transition temperature is required to be at most  $0\,^{\circ}$ C, and preferably -20 to - 130 $^{\circ}$ C. When the glass transition temperature is more than  $0\,^{\circ}$ C, an improving effect for impact resistance remarkably decreases.

[0015] Examples of the rubber satisfying these conditions are a diene rubber, an acrylic rubber, a silicon rubber and an olefin rubber, but the rubber is not limited thereto. Examples of the diene rubber are a butadiene rubber, a styrene-butadiene rubber, an acrylonitrile-butadiene rubber and the like. Examples of the acrylic rubber are a butyl acrylate rubber, a butadiene-butyl acrylate rubber, a 2-ethylhexyl acrylate-butyl acrylate rubber and the like. Examples of the silicon rubber are a polydimethyl siloxane rubber and the like. Examples of the olefin rubber are an ethylene-propylene rubber, an ethylene-propylene-diene rubber and the like.

[0016] The cross-linking agent used for rubber polymerization prevents the rubber particle from collapsing into smaller pieces at molding. When an amount thereof is too high, the collapsing at molding does not rise at all, but voids are not easily expanded at impact (under stress) and the impact strength of the final molded article is small. Examples of the cross-linkable monomer as the cross-linking agent are allyl methacrylate, divinylbenzene, diallyl phthalate, polyethylene glycol dimethacrylate, 1,3-butylene glycol dimethacrylate, ethylene glycol dimethacrylate and the like. These may be used solely or in a combination use of two or more thereof. An amount of the cross-linking agent is 0.1 to 5 parts by weight, preferably 0.1 to 1.5 parts by weight, more preferably 0.2 to 0.7 part by weight based on 100 parts by weight of the rubber component monomer. When the amount thereof is more than 5 parts by weight, voids can not be easily expanded and then an improving effect for impact resistance of the obtained molded article becomes small. On the other hand, when the amount is less than 0.1 part by weight, the rubber particles collapse into pieces to decrease the improving effect for impact resistance of the resulted molded article.

[0017] When a monomer is polymerized in the presence of a hollow rubber particle, the monomer may be polymerized as it is with the hollow rubber particle to obtain a graft polymer, or a larger graft polymer may be obtained by enhancement methods such as acid enhancement and salt enhancement of a rubber particle.

[0018] Voids of a rubber which are hollow in a latex state can be confirmed by TEM observation after embedding a rubber latex in an epoxy resin and the like, dying it with ruthenium tetraoxide and the like. Further, the void ratio can be calculated by exactly measuring the particle size of a rubber latex by Microtrac UPA (Ultrafine Particle Analyzer) and the like, and then measuring light scattering strength of the same rubber latex. The void ratio of hollow rubber particle in a latex is 3 to 90 % by volume, preferably 10 to 60 % by volume from the viewpoint of improvement in impact resistance of the final molded article. When the void ratio is more than 90 % by volume, the rubber particle may collapse at molding, and impact strength can not be stably improved. When the void ratio is less than 3 % by volume, formation

and expansion of voids in a rubber are not easily propagated at impact in the final molded article, and an improving effect for impact resistance decreases.

[0019] In order to maximize the improving effect, the particle size of the graft copolymer of the present invention is preferably 0.05 to  $2.0~\mu m$ , but the suitable value is slightly varied depending on the kind of a thermoplastic resin. Out of this range, the improving effect for impact resistance tends to decreases.

[0020] There is no particular limitation for the method for preparing hollow rubber, and the rubber can be efficiently prepared by using emulsion polymerization.

[0021] The graft copolymer of the present invention is obtained by polymerizing 10 to 70 % by weight, preferably 12 to 40 % by weight of a monomer in the presence of 30 to 90 % by weight, preferably 60 to 88 % by weight of a hollow rubber component. When the amount of the hollow rubber component is less than 30 % by weight, the improving effect for impact resistance is small. When it is more than 90 % by weight, a particle of the impact modifier collapses at molding the mixed composition, and the improving effect for impact resistance of the molded article becomes small.

**[0022]** The monomer polymerized in the presence of a hollow rubber particle is a monomer or monomers containing at least one selected from the group consisting of a (meth)acrylate compound, an aromatic vinyl compound, a vinyl cyanide compound and vinyl chloride in an amount of at least 60 % by weight and the other monomer copolymerizable therewith in an amount of at most 40 % by weight.

[0023] Examples of the (meth)acrylate are methacrylates such as methyl methacrylate, ethyl methacrylate, butyl methacrylate and 2-ethylhexyl methacrylate; acrylates such as methyl acrylate, ethyl acrylate, butyl acrylate and 2-ethylhexyl arylate; and the like. Examples of the aromatic vinyl compound are styrene, α-methylstyrene, chlorstyrene and the like. Examples of the vinyl cyanide compound are acrylonitrile, methacrylonitrile and the like.

[0024] Examples of the other copolymerizable monomer are (meth)acrylates other than the above-described esters such as glycidyl (meth)acrylate, and maleimide compounds such as maleimide and N-phenylmaleimide.

[0025] Examples of the vinyl chloride resin used in the present invention are a poly(vinyl chloride), and a copolymer of at least 50 % by weight of vinyl chloride and a monomer copolymerizable with vinyl chloride such as vinyl acetate and ethylene, a chlorinated vinyl chloride resin and the like.

[0026] An amount of the impact modifier used is usually 1 to 30 parts by weight based on 100 parts by weight of the vinyl chloride resin. When the amount is less than 1 part by weight, an improving effect for impact resistance is not recognized. When the amount is more than 30 parts by weight, there is a tendency that a tensile strength of the obtained molded article decreases.

[0027] The present invention is explained below by way of Examples in detail, but the present invention is not limited thereto. In Examples "part" and "%" mean "part by weight" and "% by weight", respectively, otherwise specified.

### **EXAMPLE 1**

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[0028] There were mixed 200 parts of water, 3.5 parts by weight of sodium oleate, 0.4 part of tripotassium phosphate, and 0.2 part by weight of a Na salt of β-naphthalenesulfonic acid -formalin condensate. After 10 parts of butyl acrylate, 3 parts of t-dodecylmercaptane, and 0.1 part of p-methane hydroperoxide were mixed, the mixture was heated to 50 °C. To the mixture was added 0.002 part of ferrous sulfate (FeSO<sub>4</sub> · 7H<sub>2</sub>O), 0.005 part of ethylenediaminetetraacetic acid · 2 Na salt, and 0.2 part of sodium formaldehyde sulfoxylate to polymerize them for 1 hour. Then, to this was continuously added a mixture of 90 parts of butyl acrylate, 27 parts of t-dodecylmercaptane and 0.9 part of p-methane hydroperoxide over 3 hours. Post polymerization was conducted for 2 hours to obtain a seed latex (S-1) having an average particle size of 0.04 μm.

[0029] There were mixed 2 part (solid content) of the seed latex S-1 and 50 parts water. Thereto was added a mixture comprising 98 parts of butyl acrylate, 0.3 part of allyl methacrylate, 0.75 part (solid content) of a 5 % sodium laurylsulfate aqueous solution and 400 parts of water, which was finely dispersed by using a homogenizer. And the mixture was stirred for 1 hour at a room temperature. To this was added 0.002 part of ferrous sulfate (FeSO<sub>4</sub>  $\cdot$  7H<sub>2</sub>O), 0.005 part of ethylenediaminetetraacetic acid  $\cdot$  2 Na salt, 0.2 part of sodium formaldehyde sulfoxylate, and 1 part of p-methane hydroperoxide. They were polymerized for 4 hours at 40 °C to obtain a hollow rubber latex (R-1) having a void ratio of 30 % by volume and a particle size of 0.15 µm.

[0030] After 85 parts (solid content) of the hollow rubber latex R-1 was heated to 45 °C, thereto were added 0.15 part (solid content) of a 5 % sodium laurylsulfate aqueous solution, 0.0016 part of ferrous sulfate (FeSO<sub>4</sub>·7H<sub>2</sub>O), 0.004 part of ethylenediaminetetraacetic acid · 2 Na salt and 0.2 part of sodium formaldehyde sulfoxylate. To this was continuously added a mixture comprising 15 parts of methyl methacrylate and 0.01 part of cumene hydroperoxide over 1 hour, and post polymerization was conducted for 1 hour to obtain a hollow graft copolymer latex (G-1) having an average particle size of 0.18 µm.

[0031] This hollow graft copolymer latex (G-1) was coagulated with calcium chloride, and subjected to thermal treatment, dehydration and drying, to obtain a hollow graft copolymer (A-1) in the form of powder.

[0032] To 100 parts of a vinyl chloride resin (available from Kaneka Corp., KANEVINYL S400) having a polymerization

degree of 400 were mixed 10 parts of the hollow graft copolymer (A-1), 3.0 parts of octyl tin mercaptide stabilizer, 1.0 part of stearyl alcohol, 0.5 part of stearic amide, 0.5 part of diol montanate, and 1.0 part of a processing aid (available from Kaneka Corp., PA10). And then, and the mixture was extruded by using a 50 mm single screw extruder (type VS50-26) made by Tanabe Plastic Machine K.K. under conditions of a rotational speed of 30 rpm, C1:140 °C, C2;160 °C, C3:160 °C, C4:165 °C and D:160 °C, to obtain a resin pellet. Then, the resin pellet was injection molded by using an injection molding machine IS-170G made by Mitsubishi Heavy Industries, Ltd. (C1: 150 °C, C2: 160 °C, C3: 170 °C, nozzle: 175 °C mold temperature: 40 °C) to obtain a test piece for dumbbell, Izod test. Izod strength was measured according to JIS-K7110. Results are shown in Table 1.

### 10 EXAMPLES 2 to 4

[0033] Preparation, coagulation, thermal treatment, dehydration, drying powder forming, compounding, molding and evaluation were conducted in the same manners as in Example 1 except that the amount of allyl methacrylate as a rubber component was changed to 0.5 part, 1.0 part and 5.0 parts, respectively. Results are shown in Table 1.

### **EXMAPLE 5**

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[0034] Preparation, coagulation, thermal treatment, dehydration, drying powder forming, compounding, molding and evaluation were conducted in the same manners as in Example 1 except that the amount of t-dodecylmercaptane at the seed polymerization was changed to 10 parts. Results are shown in Table 1.

### **COMPARATIVE EXAMPLE 1**

[0035] A mixture comprising 100 parts of butyl acrylate, 0.5 part of allyl methacrylate, 1.2 parts (solid content) of a 5 % sodium laurylsulfate aqueous solution and 450 parts of water was finely dispersed by using a homogenizer, and the mixture was stirred for 1 hour at a room temperature. To the mixture was added 0.002 part of ferrous sulfate (FeSO<sub>4</sub> · 7H<sub>2</sub>O), 0.005 part of ethylenediaminetetraacetic acid · 2 Na salt, 0.2 part of sodium formaldehyde sulfoxylate and 1.0 part of p-methane hydroperoxide. They were polymerized for 4 hours at 40 °C to obtain a usual rubber latex (R-2) having a void ratio of 0 % by volume and a particle size of 0.15 μm.

[0036] After 85 parts (solid content) of the usual rubber latex (R-2) was heated to 45 °C, thereto were added 0.15 part (solid content) of a 5% sodium laurylsulfate aqueous solution, 0.001 part of ferrous sulfate (FeSO<sub>4</sub> · 7H<sub>2</sub>O), 0.004 part of ethylenediaminetetraacetic acid · 2 Na salt, and 0.2 part of sodium formaldehyde sulfoxylate. To the mixture was continuously added a mixture of 15 parts of methyl methacrylate and 0.01 part of cumene hydroperoxide over 1 hour. Post polymerization was conducted for 1 hour to obtain a usual graft copolymer latex (G-2) having an average particle size of 0.18 µm. Post treatment, molding and evaluation were conducted in the same manner as in Example 1. Results are shown in Table 1.

### **COMPARATIVE EXAMPLE 2**

[0037] Polymerization, coagulation, thermal treatment, dehydration, drying powder forming, compounding, molding and evaluation were conducted in the same manner as in Comparative Example 1 except that the amount of allyl methacrylate as the rubber component was changed to 7.0 part. Results are shown in Table 1.

### **COMPARATIVE EXAMPLES 3 to 4**

[0038] Polymerization, coagulation, thermal treatment, dehydration, drying powder forming, compounding, molding and evaluation were conducted in the same manner as in Example 1 except that the amount of allyl methacrylate as the rubber component was changed to 7 parts and 10 parts, respectively. Results are shown in Table 1.

### 50 COMPARATIVE EXAMPLE 5

[0039] Polymerization, coagulation, thermal treatment, dehydration, drying, compounding, molding and evaluation were conducted in the same manner as in Example 1 except that the amount of the hollow rubber in the graft polymerization was changed to 5 parts and the amount of methyl methacrylate was changed to 95 parts. Results are shown in Table 1.

5		Vinyl chloride resin molded article	Izod impact Strength (kgf-cm/cm²)	51	65	44	31	. 40		21	14	. 22	16	<b>&amp;</b>
10		Vinyl chloride molded article	Amount of rubber (parts)	85	85	82	82	85	•	82	85	85	85	
20	TABLE 1	lded article	Void ratio of rubber particle (% by volume)	30	30	30	30	10		0	0	30	30	30
30		Graft copolymer in molded article	Amount of cross-linking agent in rubber (%)	0.3	0.5	1.0	5.0	0.5		0.5	7.0	7.0	10.0	0.5
35		Graf	Kind of rubber	Hollow rubber	Hollow rubber	Hollow rubber	Hollow rubber	Hollow rubber		Usual rubber	Usual rubber	Hollow rubber	Hollow rubber	Hollow
40			Ex. No.	1	7	က	4	5	Com. Ex.	No. 1	8	<b>ෆ</b>	4	S

### **INDUSTRIAL APPLICABILITY**

[0040] The impact modifier of the present invention has an excellent improving effect for impact resistance toward a vinyl chloride resin, since it uses a suitable cross linking agent and a rubber particle having a suitable void ratio.

# 50 Claims

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1. An impact modifier for a vinyl chloride resin, which is prepared by polymerizing 10 to 70 % by weight of a monomer or monomers containing 60 to 100 % by weight of at least one vinyl monomer selected from the group consisting of a (meth)acrylate compound, an aromatic vinyl compound, a vinyl cyanide compound and vinyl chloride, and 0 to 40 % by weight of the other monomer copolymerizable therewith, in the presence of 30 to 90 % by weight of a hollow rubber particle of which glass transition temperature Tg is at most 0 °C and a void ratio is 3 to 90 % by volume in a latex state, and which contains a cross linking agent in an amount of 0.1 to 5 parts by weight based on 100 parts by weight of the total rubber component monomer.

- 2. The impact modifier of Claim 1, wherein the amount of the cross linking agent is 0.1 to 1.5 parts by weight.
- 3. The impact modifier of Claim 1, wherein the amount of the cross linking agent is 0.2 to 0.7 part by weight.
- 5 4. The impact modifier of Claim 1, wherein an average particle diameter of the hollow rubber in a latex state is 50 to 300nm.
  - 5. A vinyl chloride resin composition comprising a vinyl chloride resin containing at least 50 % by weight of vinyl chloride, and the impact modifier of Claim 1.

# INTERNATIONAL SEARCH REPORT

International application No.
PCT/JP00/07220

A. CLASS	SIFICATION OF SUBJECT MATTER C1 <sup>2</sup> C08L51/00, C08L27/06, C08L	7291/00				
According to	o International Patent Classification (IPC) or to both na	ational classification an	d IPC			
B. FIELD	S SEARCHED					
Minimum de Int.	ocumentation searched (classification system followed Cl <sup>7</sup> C08L51/00, C08L27/06, C08L		ols)			
Documentat	ion searched other than minimum documentation to the	e extent that such docur	nents are included	in the fields scarched		
Electronic d	ata base consulted during the international search (nam	e of data base and, who	ere practicable, sea	rch terms used)		
C. DOCU	MENTS CONSIDERED TO BE RELEVANT					
Category*	Citation of document, with indication, where ap	· · · · · · · · · · · · · · · · · · ·		Relevant to claim No.		
X Y	JP, 10-310714, A (Kanegafuchi ( 24 November, 1998 (24.11.98), Claims; Par. Nos. [0005], [0011 Claims; Par. Nos. [0005], [0011 & WO, 98/47961, Al & EP, 9785; & CN, 1253574, A	1] 1]	, Ltd.),	1-5 1-5		
Y	JP, 04-126771, A (Japan Synthet 27 April, 1992 (27.04.92), Claims; page 3, lower left colu the bottom to lower right colur (Family: none)	umn, the 6 <sup>th</sup> l:		1-5		
Y	JP, 07-309918, A (Takeda Chemic 28 November, 1998 (28.11.98), Claims; Par. Nos. [0004] to [00		· ·	1-5		
	r documents are listed in the continuation of Box C.	y annex.				
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